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Ge N-Channel MOSFETs with ZrO₂ Dielectric Achieving Improved Mobility

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Abstract

High-mobility Ge nMOSFETs with ZrO_2 gate dielectric are demonstrated and compared against transistors with different interfacial properties of ozone (O_3) treatment, O_3 post-treatment and without O_3 treatment. It is found that with O_3 treatment, the Ge nMOSFETs with ZrO_2 dielectric having a EOT of 0.83 nm obtain a peak effective electron mobility ($\mu_{\rm eff}$) of 682 cm²/Vs, which is higher than that of the Si universal mobility at the medium inversion charge density ($Q_{\rm inv}$). On the other hand, the O_3 post-treatment with Al_2O_3 interfacial layer can provide dramatically enhanced- $\mu_{\rm eff}$, achieving about 50% $\mu_{\rm eff}$ improvement as compared to the Si universal mobility at medium $Q_{\rm inv}$ of 5 × 10¹² cm⁻². These results indicate the potential utilization of ZrO_2 dielectric in high-performance Ge nMOSFETs.

Keywords: Germanium, ZrO₂, MOSFET, CMOS, Mobility

Background

GERMANIUM (Ge) has exhibited advantages of higher carrier mobility and lower processing temperature compared with Si devices. These make Ge to be an alternative for applications of ultrascaled CMOS logic devices and thin-film transistors (TFTs) as top layer in three-dimensional integrated circuits [1–3]. In the past few years, great efforts have been focused on surface passivation, gate dielectric, and channel engineering for Ge p-channel metal—oxide—semiconductor field-effect transistors (MOSFETs), which have contributed to significant improvement in electrical performance for the p-channel devices.

But for Ge n-channel MOSFETs, low effective carrier mobility (μ_{eff}) caused by poor interfacial layer of gate stack strongly limits the performance of the devices. Various surface passivation techniques including Si passivation [1], plasma post-oxidation [4], and InAlP passivation [5] and several high- κ dielectrics including HfO₂,

 ${
m ZrO_2}$ [6–8], ${
m Y_2O_3}$ [9], and ${
m La_2O_3}$ [10] have been explored in Ge nMOSFETs to boost the electron $\mu_{\rm eff}$. It was demonstrated that ${
m ZrO_2}$ dielectric integrated with Ge channel can provide a robust interface due to that a ${
m GeO_2}$ interfacial layer can react and intermix with the ${
m ZrO_2}$ layer [7]. A decent hole $\mu_{\rm eff}$ has been reported in Ge p-channel transistors [6–8], while there is still a lot of room for improvement in electron $\mu_{\rm eff}$ for their counterparts.

In this work, Ge nMOSFETs with ${\rm ZrO}_2$ gate dielectric are fabricated to achieve improved $\mu_{\rm eff}$ over Si in the entire range of inversion charge density ($Q_{\rm inv}$). Ge transistors obtain a 50% improvement in electron $\mu_{\rm eff}$ compared to the Si universal mobility at a medium $Q_{\rm inv}$ of $5.0 \times 10^{12} {\rm \, cm}^{-2}$.

Experimental

The key process steps for fabricating Ge nMOSFETs on 4-inch p-Ge(001) wafers with a resistivity of 0.136–0.182 Ω cm are shown in Fig. 1a. The source/drain (S/D) regions were implanted with phosphorous ion at a dose of 1×10^{15} cm⁻² and an energy of 30 keV followed by dopant activation annealing at 600 °C. After the pre-gate cleaning, Ge wafers were loaded into an atomic layer deposition chamber for the formation of the gate dielectric layer(s): Al_2O_3/O_3 oxidation/ ZrO_2 , ZrO_2 , or O_3

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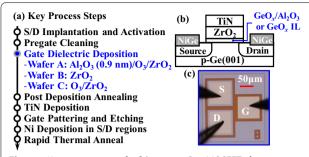


Fig. 1 a Key process steps for fabricating Ge nMOSFETs. **b** Cross-sectional schematic and **c** microscope image of the fabricated devices

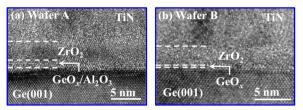


Fig. 2 HRTEM images of **a** TiN//ZrO₂/Al₂O₃/GeO_x/Ge, **b** TiN/ ZrO₂/GeO_x/Ge stacks for the devices on wafers A and B, respectively

oxidation/ZrO $_2$ for wafers A, B, or C, respectively. For wafer A, 0.9 nm Al $_2$ O $_3$ was used to protect the channel surface during O $_3$ oxidation. O $_3$ oxidation was carried out at 300 °C for 15 min for both wafers A and C. For all the wafers, the thickness of ZrO $_2$ was ~3.3 nm. Subsequently, TiN(100 nm) gate metal was deposited via physical reactive sputtering, and lithography patterning and reactive ion etching were used to form the gate electrode. After that, a 25-nm-thick Ni layer was deposited in S/D regions. Finally, the post-metallization annealing (PMA) at 350 °C for 30 s was carried out to form the Ni germanide and improve the interface quality. Schematic and microscope images of the fabricated transistor are shown in Fig. 1b, c, respectively.

Figure 2a, b shows the high-resolution transmission electron microscope (HRTEM) images of the gate stacks on wafers A and B, respectively. The unified thickness of the Al_2O_3/GeO_x interfacial layer (IL) for wafer A is ~ 1.2 nm indicating the 0.2-0.3 nm GeO_x . For the device on wafer B, an ultrathin GeO_x IL was experimentally demonstrated [7].

Results and Discussion

The measured capacitance (*C*) and the leakage current (*J*) characteristics for Ge MOS capacitors on wafers A, B, and C are measured and shown in Fig. 3a, b, respectively. The equivalent oxide thickness (EOT) of the devices

on wafers A, B, and C is extracted to be 1.79, 0.59, and 0.83 nm, respectively. Assuming the ${\rm GeO}_x$ IL provides an extra EOT of \sim 0.25 nm for wafers A and C by comparing wafers B and C, the 3.3 nm ${\rm ZrO}_2$ contributes an EOT of \sim 0.6 nm with κ value of \sim 21.8, which is consistent with the previous reported value of amorphous ${\rm ZrO}_2$ [11]. These derived results also confirm that the thickness in ${\rm GeO}_x$ IL on wafer B is negligible.

The ${\rm GeO}_x/{\rm Al}_2{\rm O}_3$ IL for wafer A and ${\rm GeO}_x$ IL for wafer C produces the EOT of ~1.2 and ~0.25 nm, respectively. The EOT of the devices can be further reduced by decreasing the IL thickness or improving the interface quality, and enhancing the permittivity of ${\rm ZrO}_2$ with some surface passivation, e.g., ${\rm NH}_3/{\rm H}_2$ plasma treatment [6]. Figure 3c compares J versus EOT characteristics for the Ge nMOSFETs in this work against values for other reported Ge devices [5, 12–17]. It is also observed that the results are consistent with the reported Ge MOS with ultra-thin EOT following the same trends, indicating the difference of leakage current shown in Fig. 3b should be mainly attributable to the difference of EOT.

Figure 4a shows measured drain current $(I_{\rm D})$ and source current $(I_{\rm S})$ versus gate voltage $(V_{\rm G})$ curves of Ge nMOSFETs from wafers A, B, and C. All transistors have a gate length $L_{\rm G}$ of 4 μ m and a gate width W of 100 μ m. The point subthreshold swing (SS), defined as ${\rm d}V_{\rm G}/{\rm d}(\log I_{\rm D})$, as a function of $I_{\rm D}$ curves for the transistors in Fig. 4a is calculated and shown in Fig. 4b. It is clarified that the transistor on wafer A exhibits the degraded $I_{\rm D}$ leakage floor and SS compared to the devices on wafers B and C. Besides the increase in EOT in devices on wafer A would bring in the increment of SS, these phenomenon should be partly attributed to the fact that the device with the ${\rm Al_2O_3}$ inserted layer has a higher density of interface traps $(D_{\rm it})$ within the bandgap of the Ge channel in comparison with the wafers B and C.

Figure 4c shows the measured output characteristics, i.e., $I_{\rm D}{-}V_{\rm D}$ curves for various values of gate overdrive $|V_{\rm G}{-}V_{\rm TH}|$ of the devices demonstrating that the Ge transistor on wafer A achieves significantly improved drive current compared to the devices on wafers B and C. Here, $V_{\rm TH}$ is defined as $V_{\rm GS}$ corresponding to an $I_{\rm D}$ of 10^{-7} A/µm. Considering the identical conditions for S/D formation, the boosted $I_{\rm DS}$ for transistors on wafer A indicates the higher $\mu_{\rm eff}$ [18–21]. The Al₂O₃ layer has not led to the degradation of $D_{\rm it}$ performance near the conduction band of the Ge channel.

Figure 5a shows the total resistance $R_{\rm tot}$ as a function of $L_{\rm G}$ for the Ge nMOSFETs with ZrO₂ dielectric with an $L_{\rm G}$ ranging from 2 to 10 μ m. The values of $R_{\rm tot}$ are extracted at a gate overdrive of 0. 6 V and a $V_{\rm D}$ of 0.05 V. The S/D resistance $R_{\rm SD}$ of the transistors is extracted to be ~13.5 k Ω μ m, utilizing the fitted lines intersecting at

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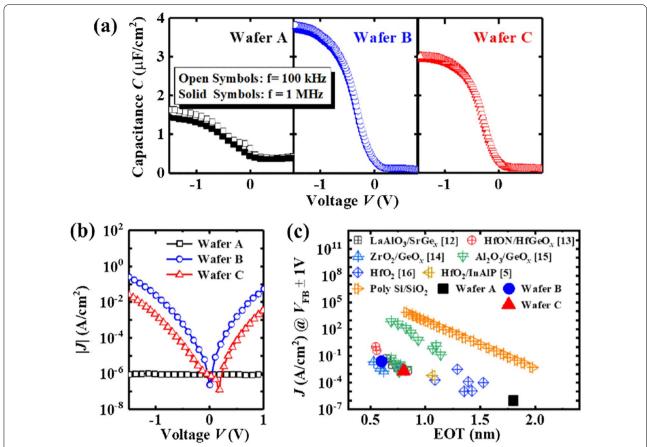


Fig. 3 a Measured C as a function of voltage V characteristics for Ge pMOS capacitors on wafers A, B, and C. **b** J versus V curves for the devices. **c** Benchmarking of J (extracted at $V_{FR} \pm 1$ V) of the Ge MOS capacitors in this work against data obtained for similar bias conditions from the literature

the *y*-axis. The similar $R_{\rm SD}$ is consistent with the identical process of PMA and S/D formation. The channel resistance $R_{\rm CH}$ values of the devices are obtained by the slope of the fitted lines, i.e., $\Delta R_{\rm tot}/\Delta L_{\rm G}$, which can be used for calculating the $\mu_{\rm eff}$ characteristics of Ge nMOSFETs. To evaluate the interface quality, interface trap densities ($D_{\rm it}$) were extracted by the following equation according to Hill's method [17]:

$$D_{\rm it} = \frac{2G_{\rm m\,max}/\omega}{qA\left[\left(\frac{G_{\rm mmax}}{\omega C_{\rm ox}}\right) + (1 - C_{\rm m}/C_{\rm ox})^2\right]}$$

where q is the electronic charge, A is the area of the capacitor, $G_{\rm m,max}$ is the maximum value of measured conductance, with its corresponding capacitance $C_{\rm m}$, ω is the angular frequency, and $C_{\rm ox}$ is gate oxide capacitance. The $D_{\rm it}$ values are calculated to be 3.7, 3.2, and $2.3\times 10^{12}~{\rm eV}^{-1}~{\rm cm}^{-2}$ for the devices on wafers A, B, and C, respectively.

It is known that the calculated values correspond to the midgap D_{it} . The device with Al_2O_3 IL on wafer A has a

higher midgap $D_{\rm it}$ compared to the devices on wafers B and C. This is consistent with the results in Figs. 3a and 4a, and the higher midgap $D_{\rm it}$ gives rise to a larger depletion capacitance dispersion in wafer A causing a higher leakage current of $I_{\rm DS}$ in comparison with the other two wafers. Note the wafer A should have the lower $D_{\rm it}$ near the conduction bandgap due to its higher $\mu_{\rm eff}$ over wafers B and C.

It is well known that $\mu_{\rm eff}$ is the bottleneck for high drive current and transconductance in Ge nMOSFETs. Here, $\mu_{\rm eff}$ can be calculated by $\mu_{\rm eff}=1/[WQ_{\rm inv}(\Delta R_{\rm tot}/\Delta L_{\rm G})]$, where $\Delta R_{\rm tot}/\Delta L_{\rm G}$ is the slope of the $R_{\rm tot}$ versus $L_{\rm G}$ as shown in Fig. 5a. $Q_{\rm inv}$ can be obtained by integrating the measured $C_{\rm inv}-V_{\rm G}$ curves. In Fig. 5b, we compare the $\mu_{\rm eff}$ versus $Q_{\rm inv}$ of the Ge nMOSFETs on wafers A, B, and C with those reported previously in [18, 22–25]. The extracted peak $\mu_{\rm eff}$ values of the transistors on wafers A and C are 795 and 682 cm²/V s, respectively, and that of Ge nMOSFETs on wafer B is 433 cm²/V s. Ge nMOSFETs with ${\rm Al}_2{\rm O}_3$ IL achieve a significantly improved $\mu_{\rm eff}$ in comparison with the transistors on wafer B or C, the devices

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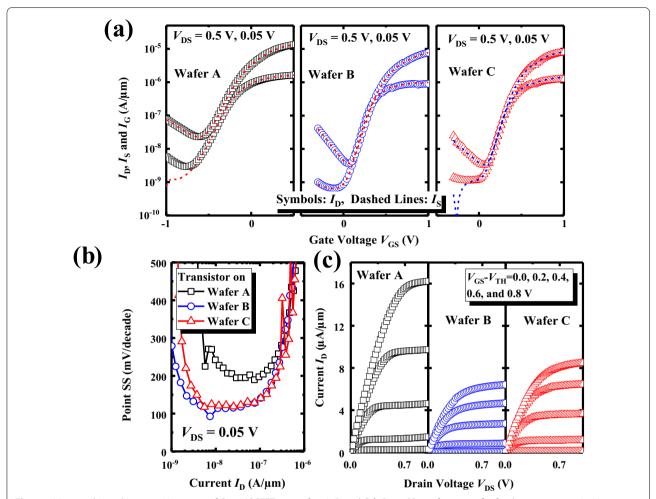


Fig. 4 a Measured I_D and I_S versus V_{GS} curves of Ge nMOSFETs on wafers A, B, and C. **b** Point SS as a function of I_D for the transistors. **c** $I_D - V_D$ characteristics show that the Ge nMOSFET on wafer A has a higher drive current compared to the devices on wafers B and C

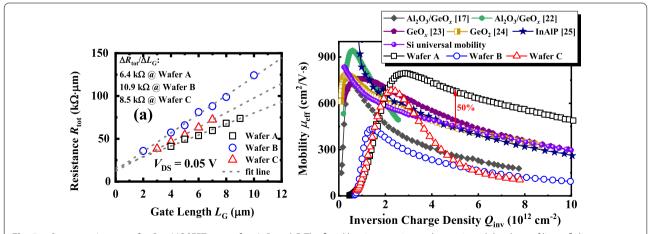


Fig. 5 a R_{tot} versus L_{G} curves for Ge nMOSFETs on wafers A, B, and C. The fitted line intersecting at the y-axis and the slope of linear fit lines are utilized to extract the R_{SD} and R_{CH} , respectively. **b** μ_{eff} for the Ge nMOSFETs in this work versus previously published results for unstrained Ge transistors. The devices on wafer A show the improved μ_{eff} than the Si universal mobility in the entire range of Q_{inv}

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in [18, 22-25] in a high field, and Si universal mobility in the entire $Q_{\rm inv}$ range. At a $Q_{\rm inv}$ of 5×10^{12} cm⁻², a 50% $\mu_{\rm eff}$ enhancement is achieved in devices on wafer A as compared to the Si universal mobility. This demonstrates that by protecting the channel surface for preventing the intermixing of ZrO₂ and GeO_x using Al₂O₃, a high-quality interface between gate insulator and Ge is realized to boost the mobility characteristics, which is also reported in the previous studies of Ge MOSFETs with ultrathin EOT [26]. μ_{eff} in transistors on wafer C is higher than the Si universal at a Q_{inv} of 2.5×10^{12} cm⁻², although it rapidly decays with the increase in Q_{inv} range. This indicates that the used O₃ oxidation before ZrO₂ deposition would improve the interfacial quality to some extent; however, it does not lead to enough flat channel surface to effectively suppress the surface roughness scattering of the carrier at high Q_{inv} due to the intermixing of ZrO_2 and GeO_r since it is reported that the generation of oxygen vacancies during the intermixing would increase the short-range order (SRO) roughness [27]. Optimizing the O₃ oxidation process or reducing the Al₂O₃ IL thickness can make the Ge transistor achieve a reduced EOT while maintaining a higher μ_{eff} at the high Q_{inv} .

Conclusions

The impacts of gate dielectric structure and morphology on Ge nMOSFET electrical characteristics are investigated. An Al $_2$ O $_3$ /ZrO $_2$ gate dielectric provides for significantly-improved $\mu_{\rm eff}$ as compared to the Si universal mobility. $\mu_{\rm eff}$ can be improved by inserting an Al $_2$ O $_3$ layer between the ZrO $_2$ and Ge channel, which, however, inevitably leads to a larger EOT. Al $_2$ O $_3$ -free Ge nMOSFETs with O $_3$ oxidation of the Ge surface prior to ZrO $_2$ deposition achieve a peak $\mu_{\rm eff}$ of 682 cm 2 /V s which is higher than that of Si at the similar $Q_{\rm inv}$.

Abbreviations

Ge: Germanium; ZrO $_2$: Zirconium dioxide; Al $_2$ O $_3$: Aluminum oxide; O $_3$: Ozone; Si: Silicon; PMA: Post-metal annealing; PDA: Post-deposition annealing; IL: Interfacial layer; TiN: Titanium nitride; MOSFETs: Metal-oxide-semiconductor field-effect transistors; ALD: Atomic layer deposition; HF: Hydrofluoric acid; $\mu_{\rm eff}$: Effective carrier mobility; PPO: Plasma post-oxidation; SS: Subthreshold swing; CET: Capacitance-equivalent thickness; EOT: Equivalent oxide thickness; Qinv: Inversion charge density; HRTEM: High-resolution transmission electron microscope; Ni: Nickel; GeO $_{x}$: Germanium oxide; $l_{\rm DS}$: Drain current; $V_{\rm GS}$: Gate voltage; $V_{\rm TH}$: Threshold voltage.

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Authors' contributions

LLC carried out the experiments and drafted the manuscript. HL and YP provided the discussion on the results. XY provided the discussion and revised the manuscript. GQH and YL supported the study and helped to revise the manuscript. YH provided constructive advice in the drafting. All the authors read and approved the final manuscript.

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Availability of Data and Materials

The datasets supporting the conclusions of this article are included in the article.

Declaration

Competing interests

The authors declare that they have no competing interests.

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